

Accelerated hygrothermal ageing of bond in FRP-masonry systems

Bahman Ghiassi¹, Paulo B. Lourenço², Daniel V. Oliveira³

ISISE, University of Minho, Department of Civil Engineering, Guimarães, Portugal

ABSTRACT

This paper addresses the results of accelerated hygrothermal (coupled temperature and moisture) tests on FRP-strengthened clay bricks aimed at investigating bond degradation mechanisms. The exposures are selected to simulate different environmental conditions and the bond degradation is periodically investigated by visual inspection and by conventional single-lap shear bond tests. The changes in the properties of material constituents have also been monitored and the results are presented and critically discussed. A decay model is then adopted for simulating the observed degradation in the specimens. The model, once validated, is used for long-term performance prediction of FRP-masonry systems and the results are compared with the environmental reduction factors proposed by available design guidelines.

Keywords: FRP; masonry; bond; durability; hygrothermal; accelerated ageing.

¹Postdoctoral Researcher, ISISE, University of Minho, Department of Civil Engineering, Azurém, 4800-058 Guimarães, Portugal. Phone: +351 253 510 499, fax: +351 253 510 217, E-mail: bahmanghiassi@civil.uminho.pt

² Professor, ISISE, University of Minho, Department of Civil Engineering, Azurém, 4800-058 Guimarães, Portugal. Phone: +351 253 510 209, fax: +351 253 510 217, E-mail: pbl@civil.uminho.pt

³Associate Professor, ISISE, University of Minho, Department of Civil Engineering, Azurém, 4800-058 Guimarães, Portugal. Phone: +351 253 510 247, fax: +351 253 510 217, E-mail: danvco@civil.uminho.pt

17 **Introduction**

18 Modern composite materials such as fiber reinforced polymers (FRPs) have been accepted as
19 effective strengthening materials for civil engineering structures. FRPs provide several
20 advantages comparing to conventional strengthening techniques which have made them
21 interesting for strengthening purposes. FRPs have also received an extensive attention in the last
22 decades for external strengthening of masonry structures (Hollaway 2010).

23 In external strengthening techniques with composite materials, the efficacy and reliability
24 of the strengthening depends intrinsically on the bond between the composite material and the
25 substrate. The bond behavior has been extensively studied in FRP-concrete systems, but in case
26 of FRP-masonry it has only recently received attention (Grande et al. 2008, Grande et al. 2011,
27 Fedele and Milani 2012, Ghiassi et al. 2012, Valluzzi et al. 2012). However, the durability and
28 long-term performance of bond still remains a challenge for masonry and concrete substrates
29 (Karbhari et al. 2003, Wu et al. 2010). Available information regarding the durability of bond
30 behavior are mostly devoted to FRP-concrete systems under aggressive environments or
31 moisture conditions, see e.g. (Karbhari and Ghosh 2009, Benzarti et al. 2010, Tuakta and
32 Buyukozturk 2011, Marouni et al. 2012, Silva et al. 2013, Kim et al. 2014), and only few
33 researches can be found regarding the FRP-masonry, see e.g. (Sciolti et al. 2012, Ghiassi et al.
34 2013a, Ghiassi et al. 2013b).

35 Structures are exposed to environmental changes or degrading agents, such as large
36 temperature and moisture variations or alkaline agents, during their service life. These changes
37 can affect the performance of the structure to a large extent which should be taken into account
38 during the design procedure or should be defeated with innovative solutions. It is thus necessary

39 to clearly understand the environmental degradation mechanisms and their effects on the
40 structural components or strengthening material.

41 Most of the environmental factors and deterioration processes are dependent on or
42 coupled with moisture and temperature, and therefore a good understanding of their effects on
43 deterioration of bond is a key step in durability modeling of FRP-strengthened masonry
44 elements. The moisture is known to play an important role in durability of bond in FRP
45 applications, as it reduces the bond strength and fracture energy (Ouyang and Wan 2008, Lau
46 and Buyukozturk 2010, Sciolti et al. 2012, Böer et al. 2013, Ghiassi et al. 2013). The degrading
47 effect of moisture is due to extensive moisture plasticization of the polymer adhesive (which
48 leads to mechanical degradation) and additional breakage of interfacial bonds (Wan et al. 2006).
49 Moreover, the moisture induced vapor and osmotic pressure in the interface can lead to local
50 debonding (Ouyang and Wan 2009). However, the degrading effect of moisture on the bond
51 behavior varies with material properties, surface treatments, and specimens configurations
52 (Sciolti et al. 2012). Temperature cycles below the epoxy glass transition temperature may cause
53 degradation in the bond due to the imposed thermal fatigue and thermal incompatibility between
54 FRP and the substrate (Karbhari et al. 2003). Furthermore, exposure to subzero temperatures and
55 freeze-thaw cycles cause degradation in the bond behavior (Silva et al. 2013). Still, the combined
56 effect of temperature cycles and moisture, the so-called hygrothermal ageing, is not known.

57 This paper addresses the results of accelerated hygrothermal (coupled temperature and
58 moisture) tests on FRP-strengthened masonry specimens aimed at investigating the bond
59 degradation in these systems. The specimens consist of GFRP-strengthened bricks prepared
60 following the wet lay-up procedure. The bond degradation is assessed by performing
61 conventional single-lap shear bond tests at different periods of exposure. The changes in

62 mechanical properties of the material constituents are also investigated. The main observations
63 and experimental results are presented and critically discussed. Finally, a decay degradation
64 model is fitted to the experimental results and is used for simulating the long-term behavior of
65 bond in FRP-strengthened masonry.

66

67 **Experimental program**

68 The experimental program addresses an investigation on the degradation of bond behavior in
69 FRP-strengthened masonry due to hygrothermal conditions. The changes in the material
70 mechanical properties and the bond behavior with exposure time are monitored periodically by
71 performing qualitative and quantitative laboratory tests. The material characterization tests,
72 specimens' preparation, accelerated exposure conditions and post-ageing test methods are
73 presented in this section.

74

75 **Materials and specimens**

76 Solid clay bricks with dimensions of $200 \times 100 \times 50 \text{ mm}^3$ are used in this study as substrate. The
77 bricks were produced by extrusion without the application of any finishing or glazing on the
78 surface. Glass Fiber Reinforced Polymer (GFRP) is used as the composite material for external
79 strengthening of the bricks. GFRP composites, compared with other conventional FRP materials,
80 have lower axial stiffness that makes them more suitable for masonry structures (Oliveira et al.
81 2011). The GFRP composites are prepared with a commercially available unidirectional E-glass
82 fiber and a compatible two-part epoxy resin as matrix, following the wet lay-up procedure. A
83 two-part epoxy primer is also used for preparation of the bricks' surfaces before application of
84 FRP composite.

85 For the material characterization tests, cubic brick specimens, dog-bone shape epoxy and
86 primer specimens and GFRP coupons are prepared according to relevant test standards, see Fig.
87 1. For the bond characterization, specimens are prepared following the wet lay-up procedure, see
88 Fig. 2. The GFRP sheets with 50 mm width are applied to 150 mm length of the brick's surface
89 leaving a 40 mm unbonded length at the loaded end. The bricks were dried in the oven before
90 application of the GFRP sheets. After cleaning the brick surface, a two-part epoxy primer is
91 applied to the brick's surface. Finally, a two-part epoxy resin is used as the matrix of the
92 composite material and also for adhesion to the masonry substrate.

93

94 **Mechanical characterization**

95 Mechanical characterization tests are performed according to relevant test standards and the
96 results are presented as the mean value of five tested specimens, see Table 1. The tests are also
97 performed on conditioned specimens for monitoring the changes of mechanical properties of
98 materials with exposure time.

99 The mechanical properties of bricks are obtained according to standards EN 772-1 (2002)
100 and EN 8942-3 (1986) in terms of compressive strength, f_{cb} and flexural tensile strength, f_{tb} . The
101 compressive strength is obtained by performing compressive tests on 40 mm height brick cubes,
102 in the flatwise direction with a 50 kN Lloyd testing machine, see Fig. 3(a). Three point bending
103 tests are performed on $160 \times 40 \times 40 \text{ mm}^3$ brick prisms to obtain the flexural tensile strength.

104 Tensile strength and elastic modulus of the epoxy resin and primer are determined from
105 tensile tests on dog-bone shape specimens, see Fig. 1. Although seven days are proposed for
106 curing the epoxy resin in the technical datasheet provided by the manufacturer, the specimens are
107 previously cured for 60 days at room temperature. Previous studies have shown that high curing

108 times are necessary for complete curing of cold-cured epoxy resins (Frigione et al. 2006, Aiello
109 et al. 2006, Sciolti et al. 2012). The specimens' preparation and tensile tests are conducted
110 following ISO 527-1 (2012). The tests are carried out with an Instron testing machine at a
111 displacement rate of 0.01 mm/min, see Fig. 3(b). Deformation of the specimens is monitored by
112 a clip gauge placed on the middle of the specimens.

113 The glass transition temperature (T_g) of the epoxy resin is also obtained by means of DSC
114 (Differential Scanning Calorimetry) test. The thermal scans are carried out between 5°C and
115 200°C with a heating rate of 10°C/min. The T_g is calculated as the mean value of four tests.
116 There are several method of obtaining T_g commonly used by researchers (Ratna, 2009) (Ratna,
117 2009) such as Differential Scanning Calorimetry (DSC), Thermo Mechanical Analysis (TMA)
118 and Dynamic Mechanical Analysis (DMA). In DSC, TMA and DMA the specimens are heated
119 through a temperature range to obtain the T_g . The results obtained from each technique are,
120 usually, different. In reality, the glass transition is a not a specific temperature, but rather a
121 temperature range where several material properties undergo a change. DSC has been widely
122 used by other researchers for durability tests, see e.g. (Sciolti et al. 2012) and is used in this
123 study. It is well known that the values of T_g depend on several factors such as heating rate and
124 rate of cooling of samples prior to measurements. Therefore, the tests in this study are performed
125 based on the procedure reported in (Mazurin & Gankin, 2007).

126 Regarding the composite materials, the specimens' preparation and mechanical tests are
127 conducted according to ISO 527-1 (2012). The mechanical properties are obtained in terms of
128 tensile strength, f_{tf} , and elastic modulus, E_f . The tests are carried out with an Instron testing
129 machine at a displacement rate of 0.01 mm/min, see Fig. 3(c). The GFRP coupons are prepared
130 following the wet lay-up procedure according to the code specifications. Throughout this study,

131 changes in the tensile strength and elastic modulus of GFRP coupons are normalized to the
132 specimens' thickness in accordance with ASTM D7565-10 (2010). In wet lay-up procedures the
133 specimens' thickness varies and the normalization of the mechanical properties by the thickness
134 can provide an accurate baseline for comparison (Cromwell et al. 2011).

135 As it can be seen in Table 1, the brick presents very low CoVs (about 4%). The GFRP
136 coupons exhibit CoVs about 15%, whereas the epoxy resin and primer exhibit intermediate
137 values, with CoVs about 10%.

138

139 **Bond characterization**

140 Single-lap shear bond tests are performed to investigate the bond behavior in the reference and
141 aged specimens. The bond tests are performed using a closed-loop servo-controlled testing
142 machine with a maximum load capacity of 50 kN. A rigid steel frame is used to support the
143 specimens appropriately and avoid misalignments in the load application. The specimens are
144 placed on the steel frame and firmly clamped to it as shown in Fig. 4. The tests are driven under
145 displacement control conditions with reference to the LVDT sensor placed at the loaded end of
146 the FRP composite. The specimens are pulled monotonically at a rate of 5 μ m/sec. The resulting
147 force, F, is measured by means of a load cell. The relative slip between the GFRP and the brick
148 is measured with the LVDTs placed along the bonded length. In general, two LVDTs are glued
149 at the loaded end (denoted by TL and TR), two in the middle of the bonded length (denoted by
150 ML and MR), and one at the free end of the FRP sheet (denoted by B).

151

152 **Hygrothermal exposure**

153 The specimens are exposed to two different hygrothermal (coupled temperature and moisture)
154 conditions in a climatic chamber. The exposures consisted of 6 hours temperature cycles from
155 +10°C to +50°C and constant relative humidity of 90% (exposure HT1) and 60% (exposure
156 HT2), see Fig. 5. In each cycle, the temperature is kept constant at +10°C for 2 hours,
157 subsequently increased to +50°C in 1 hour, followed by 2 hours constant temperature at +50°C.
158 Then, the temperature is decreased again to +10°C in 1 hour resulting in 6 hours cycles of
159 exposure. The specimens are subjected to a total of 225 cycles of HT1 and 820 cycles of HT2
160 conditions. The difference in exposure period is due to the fact that the climatic chamber was
161 available for limited periods of time.

162 As stated in introduction, the available literature on hygrothermal exposure or cyclic
163 temperature exposure conditions on FRP-bonded components is rare and not standardized.
164 Among the few studies found, different exposure conditions are chosen. On the other hand, most
165 of the temperature cycles studies are limited to freeze-thaw conditions and cycle of temperatures
166 in the positive range combined with relative humidity is not common. The temperature cycles
167 used in this study are therefore selected as a reference for further durability tests while considering
168 several factors. The +50°C is relatively a high temperature and is chosen to accelerate the
169 degradation phenomenon, while being far enough from the epoxy resin T_g (70°C). Since
170 environmental conditions can cause reduction (or increment) of T_g in the epoxy resin, the
171 maximum temperature in the thermal cycles should avoid reaching the T_g of the epoxy resin
172 during the tests (Karbhari 2007). It should be noted that measurement of changes in T_g during the
173 tests is critical to understand the state of degradation in the epoxy resin and whether the
174 environmental temperature exceeds this value or not. In this study, the T_g is measured only for
175 the un-conditioned specimens.

176

177 **Post-ageing tests**

178 Post-ageing tests consist of mechanical characterization and single-lap shear bond tests on the
179 specimens at different exposure periods. The specimens are taken from the climatic chamber and
180 then stabilized in laboratory conditions for four days, before performing the post-ageing tests.
181 Five specimens are tested in each exposure period and the average results are presented next.
182 The mechanical tests are performed on brick cubes, epoxy specimens and GFRP coupons as
183 explained in sec 2.2. Meanwhile, single-lap shear bond tests are performed to investigate the
184 degradation of bond behavior, as explained in sec. 2.3.

185

186 **Results and discussion**

187 **Material properties**

188 The changes in the compressive strength of bricks due to the hygrothermal exposures are shown
189 in Fig. 6. The change in the bricks compressive strength is negligible in all exposure conditions
190 with a low CoV (maximum 10%). The results show the good resistance of the bricks to the
191 environmental exposures considered in this study.

192 The changes in mechanical properties of epoxy resin, namely elastic modulus and tensile
193 strength, are presented in Fig. 7 and Fig. 8. A similar degradation is observed in elastic modulus
194 and tensile strength. Some fluctuations can be observed in the test results with exposure time.
195 This can be due to several factors including scatter in the experimental results as a nature of
196 experimental testing, differences in the microstructure and curing of the specimens (although made
197 using the same procedures in a single batch), differences in the porosity of the specimens and
198 variation of the material properties. As these fluctuations are observed between specimens exposed to
199 different exposure periods, the global degradation trend is more important than point-to-point

200 comparison. While the latter is investigated in the last section of this paper and the graphs are
201 presented together with the predictive decay models, the former is also addressed next at some
202 critical points for performing a local comparison between both exposure conditions. For the
203 elastic modulus, the degradation after 225 cycles is 7%, for both exposures. Meanwhile, for the
204 tensile strength, HT1 induced 20% reduction after 225 cycles of exposure being two times more
205 than the corresponding degradation due to HT2 (10% reduction in both exposures). This
206 difference is clearly due to the moisture attack in exposure HT1 which has resulted in higher
207 degradation in the specimens. The total observed degradation in the epoxy tensile strength is
208 14% (at 820 cycles) in HT2. 225 cycles is chosen for point-to-point comparison between both
209 exposures at the end of exposure HT1 to avoid extrapolation of the data. The results show that
210 the epoxy resin used in this study has less durability in high humid environments (exposure
211 HT1), although longer cycles of exposure are needed for a clear conclusion. The CoVs of the
212 tests in all exposures are in the range of 2% to 13% which seem reasonable for testing material
213 properties (Haldar and Mahadevan 2000).

214 The changes in mechanical properties of GFRP coupons together with the scatter of the
215 experimental results are presented in Fig. 9 and Fig. 10. Again, a relatively similar degradation
216 trend is observed in elastic modulus and tensile strength, with exposure HT1 inducing higher
217 degradation in the specimens, as expected. The elastic modulus and tensile strength of the GFRP
218 decreased 23% and 22%, respectively, after 225 cycles of HT1 exposure, with corresponding
219 reductions of 9% and 13% in HT2. The total observed reduction for the elastic modulus and the
220 tensile strength was 22% and 13% in HT2 exposure showing that the degradation in HT2
221 exposure has reached a residual value. However, reaching a residual value in HT1 cannot be
222 easily concluded at this stage. The CoVs of the experimental results in both exposures are in the

223 range of 5% to 12% which are again typical for testing material properties (Haldar and
224 Mahadevan 2000).

225 The observed degradation can be attributed to different degrading mechanisms. Exposure
226 to temperature cycles, besides the above mentioned effects on the matrix, may cause interfacial
227 micro-cracking due to the difference between thermal expansion coefficients of glass fibers and
228 epoxy resin (Dutta and Hui 1996, Karbhari 2002). The thermal expansion coefficient of E-glass
229 fibers is around $5 \times 10^{-6}/^{\circ}\text{C}$, while for the epoxy resin is in the range of $3 \sim 5 \times 10^{-5}/^{\circ}\text{C}$ (CNR-DT200
230 2004). This one-order magnitude difference of thermal expansion coefficient produces large
231 interfacial thermal stresses at the fiber/epoxy interfaces. In conclusion, the observed degradation
232 in the specimens in HT2 conditions can be a combination of epoxy post-curing, induced thermal
233 fatigue, and the thermal mismatch between epoxy resin and glass fibers. In wet environments
234 (HT1), GFRP coupons absorb moisture which causes degradation in the epoxy resin properties,
235 as described before. Moreover, the water attacks glass fibers resulting in degradation of their
236 mechanical properties and surface energy (Schutte 1994). The fiber/epoxy interface may also
237 degrade due to the degradation of fiber and epoxy resin and also the produced osmotic pressure
238 at the interface (Karbhari 2007).

239

240 **Bond behavior**

241 *Visual inspection*

242 All the specimens are visually inspected periodically, before performing the debonding tests, for
243 investigating the existence of visible interfacial damage or FRP delamination. Although due to
244 the transparency of the epoxy resin, FRP delamination is observable with visual inspection, IR

245 thermography tests are also performed on specimens for better characterization of delaminations.
246 The results of the IR thermography tests are presented in (Ghiassi et al. 2014).

247 In general, progressive FRP delamination is observed in the specimens as the exposure
248 cycles increased. The size of delamination is characterized from the IR photos, see Fig. 11(a),
249 with the aim of adopted quantitative IR analysis as explained in (Ghiassi et al. 2014). The
250 delaminations, being at the FRP/brick interface, are generally larger in the specimens subjected
251 to HT1 cycles. The average equivalent debonding length growth with exposure cycles is plotted
252 in Fig. 11(b). The equivalent debonding length is obtained as the debonded area divided by FRP
253 width. This parameter, while providing a clear idea of the debonding progress, is useful for
254 numerical simulations when two-dimensional models are adopted (as is the case for most
255 situations). The specimens exposed to HT2 conditions show a linear debonding growth with a
256 relatively slow rate. However, the debonding growth in the specimens exposed to HT1
257 conditions is rather large, with an exponential incremental rate.

258 The observed delamination in the specimens can be attributed to the thermal
259 incompatibility between the composite material and the brick used in this study, as explained
260 before. Additionally, cyclic temperature conditions produce thermal fatigue and may cause FRP
261 delamination from the brick surface during the environmental exposures. The effect of moisture
262 presence on the debonding growth rate is clear in exposure HT1. The moisture attack has
263 resulted in the reduction of surface energy at the FRP-brick interface and therefore the interfacial
264 thermal stresses induced larger delaminations in the specimens exposed to HT1 conditions.

265

266 *Delamination tests*

267 The changes in the debonding force and slip of the specimens is presented in Fig. 12 and Fig. 13
268 for both exposure conditions. The debonding force has progressively decreased with the number
269 of exposure cycles. The debonding force decreased 45% and 20% after 225 cycles of HT1 and
270 HT2, respectively. The average reduction of debonding force is 13% at the end of HT2
271 exposures. Again, it seems that the degradation has reached a residual value. Moreover, the
272 debonding behavior changed from a brittle failure mode to a progressive and less brittle failure
273 mode in exposure HT1. Similar changes in the bond behavior have also been reported in the
274 literature for the specimens exposed to freeze-thaw and wet-dry cycles, see e.g. (Davalos et al.
275 2008). The higher degradation observed in specimens exposed to HT1 is due to the moisture
276 attack to the interfacial bond between FRP composite and brick and also to the constituent
277 materials. The debonding slip, presented in Fig. 13, is the slip of the GFRP at the moment of
278 debonding obtained from the LVDTs measurements. The debonding slip has been reduced with
279 exposure time in both exposures with a relatively high CoV. It seems that, in HT2 exposure, the
280 reduction of the debonding slip has reached a residual value.

281 Regarding the failure mode, a progressive change of failure mode from cohesive to
282 adhesive is observed in the specimens after HT1 exposure, see Fig. 14. However, no specific
283 change of failure mode is observed in the specimens exposed to HT2 conditions. Such a change
284 in the failure mode, also reported in (Green et al. 2000), can be attributed to the observed bond
285 degradation during hygrothermal exposure. It seems that the moisture attack has produced a
286 weak line at the FRP-masonry interface (by reducing the interfacial fracture energy) which has
287 resulted in the observed change of failure mode.

288 A drawback of strength-based approaches in investigating the environmental effects on
289 the bond behavior, as done in Fig. 12, is that the results depend on the geometrical characteristics
290 of the specimens. Fracture mechanics approaches seem to be more appropriate in debonding
291 problems (Tuakta and Buyukozturk 2011). In fracture-based approaches, the degradation
292 parameter is usually the fracture energy or the critical energy release rate. According to CNR
293 DT200 (2004), the bond fracture energy can be obtained from the debonding tests as:

$$294 \quad G_f = \frac{P_{\max}^2}{b_f^2(2E_f t_f)} \quad (1)$$

295 where, P_{\max} is the debonding strength, b_f is the FRP width, E_f is the FRP elastic modulus, and
296 t_f is the FRP thickness. It should be noted that this equation is correct if the bonded length is
297 more than the effective bond length, which is assumed to be the case in this study throughout the
298 whole exposure period. Measurement of the changes in the effective bond length is possible by
299 using strain gauges or through using advanced full field measurement techniques and is not
300 performed in this study. The changes of the bond fracture energy with exposure time are
301 obtained using Eq. (1) and the results are presented in Fig. 15. The results are presented as
302 normalized to the reference fracture energy. The average bond fracture energy value reaches
303 0.45 N/mm for the reference specimens. The fracture energy has moderate changes due to HT2
304 conditions (with a 20% reduction at the end of exposure), while a large degradation is observed
305 in the specimens exposed to HT1 conditions (60% total reduction at the end of exposure). It
306 seems that the interfacial fracture energy has reached a residual value in exposure HT2. A
307 comparison between HT1 and HT2 exposures shows that the moisture affects the interfacial
308 fracture energy to a large extent. The interfacial fracture energy in exposure HT1 may have

309 reached a residual value as well, although performing longer exposure periods is necessary for a
310 clear conclusion.

311 Assuming that the FRP width and thickness are constant during the hygrothermal
312 exposures, the debonding force is directly related to the square root of interfacial fracture energy
313 and FRP elastic modulus, see Eq. (1). Therefore, the participation of each factor in the
314 degradation of the debonding force can be obtained by plotting the changes in the normalized
315 square root of fracture energy and FRP elastic modulus as shown in Fig. 16. In exposure HT2,
316 the degradation of fracture energy and FRP stiffness has similar effects on the global bond
317 behavior, with the fracture energy having a larger contribution in the bond degradation. The
318 interfacial bond degradation can be attributed to the existing thermal incompatibility inside the
319 composite system and FRP-brick interface. However, when moisture exists in the environment,
320 such as for exposure HT1, the interfacial degradation of the bond has a major effect on the global
321 bond degradation when compared to the FRP elastic modulus. This was expected as moisture is
322 known to cause degradation in the bond strength and fracture energy (Ouyang and Wan 2008,
323 Lau and Buyukozturk 2010, Sciolti et al. 2012, Ghiassi et al. 2013a).

324

325 **Long-term predictions**

326 This section presents the application of a decay model in predicting the observed degradation in
327 mechanical properties of materials and bond behavior. It should be noted that using predictive
328 models in accelerated ageing tests requires a deep knowledge of the active degradation
329 mechanisms and a large experimental database. The experimental results presented here
330 demonstrated the need for performing longer accelerated ageing tests for exposure HT1 and
331 therefore the predictions made are limited to the available data. Even though, the use of

332 predictive models assists in better understanding the degradation trends, allows a first
333 comparative study between different exposure conditions and also contributes to development of
334 constitutive models for numerical modeling approaches. Once the decay models are fitted to
335 experimental results, they are used for simulating the long-term performance of the FRP-
336 strengthened elements.

337

338 **Degradation modeling**

339 An exponential decay model proposed by Phani and Bose (1986) is used next, see Eq. (2). This
340 model has been previously used by other authors for predicting the materials degradation under
341 moisture and temperature conditions, see e.g. (Chen et al. 2006, Nguyen et al. 2012).

$$342 \quad \sigma_t = \sigma_\infty + (\sigma_0 - \sigma_\infty) \exp(kt) \quad (2)$$

343 where, σ_t is the strength after exposure to environmental conditions for a period of t , σ_0 is the
344 unconditioned strength, σ_∞ is the residual strength after complete degradation and k is the rate
345 of degradation. Here, the parameters are directly obtained by performing a regression analysis on
346 the experimental data and the results are presented in Fig. 17 to Fig. 18. For the HT2 exposure,
347 the regression analysis is performed for the first 300 cycles so that the accuracy of the model in
348 predicting the degradation until the end of the tests (820 cycles) can be evaluated.

349 The percent error in the predictions for each exposure is presented in Table 2 and Table
350 3. Here the f_{te0} , E_{te0} , f_{tf0} , E_{tf0} , P_0 and G_{f0} are the epoxy tensile strength and elastic modulus, GFRP
351 tensile strength and elastic modulus, debonding force and fracture energy of the un-conditioned
352 specimens, respectively. The accuracy of the models is relatively good for all mechanical
353 properties. For the epoxy tensile strength the error range is up to 8.8%, while the error for the
354 elastic modulus is in up to 7.6%. For the GFRP coupons, the error range is up to 4.8% and 10.5%

355 for the tensile strength and elastic modulus, respectively. Meanwhile, the error in prediction of
356 the debonding force is 18.2% and for the fracture energy is up to 46.4%. The reasonable
357 accuracy of the models in HT2 exposure until the end of the tests, although fitted with 300 cycles
358 of experimental data, shows the suitability of the adopted regression method.

359

360 **Long-term performance modeling**

361 The proposed predictive models for HT1 and HT2 exposures are respectively used for long-term
362 performance assessment of bond and material properties in environments with high and average
363 relative humidity. As stated before, establishing a link between real exposure conditions and
364 accelerated ageing tests is a complicated task which requires extensive experimental tests. The
365 number of cycles experienced by the materials is considerably influenced by geographic
366 location. Some authors have tried to simulate the real condition of freeze-thaw cycles in different
367 regions assuming each year is equal to 30 to 50 cycles (Barnes 1990, Soudki and Green 1997,
368 Lesko 1999). As an average, it is assumed here that each 40 cycles of hygrothermal exposures
369 represent 1 year life of the structure in real exposure conditions.

370 Assuming that each 40 cycles represents 1 year of real exposure conditions, the
371 estimations are made for 2000 cycles of HT1 and HT2 exposure for high and average relative
372 humidity environments, respectively. The model reaches a residual value after 50 years (200
373 cycles), which is the standard code value for structural life expectancy. After 50 years, the
374 degradation in the tensile strength of epoxy resin is 25% and 10% for wet and average humidity
375 environmental conditions, respectively. These values are 7% and 18% for the elastic modulus.
376 For the GFRP, the degradation is 24% and 14% in case of tensile strength, meanwhile it is 26%
377 and 21% for the elastic modulus. For the bond strength 68% and 21% degradation is predicted

378 for wet and average humidity environmental conditions, respectively. The corresponding
379 predictions for the bond fracture energy are 80% and 25% reductions. The effect of moisture in
380 high relative humidity environments is clear in the predicted degradations.

381 The lack of knowledge on the long-term performance of FRP-bonded systems has
382 become a major challenge for engineers at the design stage. Some design guidelines, see e.g.
383 (CNR-DT200 2004, ACI 440.7R-10 2010), have recently implemented reduction factors on the
384 material properties for simulating the environmental deterioration effects. According to CNR DT
385 200 (2004), 25%, 35% and 50% degradation should be assumed in FRP-bonded systems
386 respectively in internal, external and aggressive exposure conditions. The long-term predictions
387 in this study show that the current design methodology may underestimate the degradation
388 factors in wet environments. The degradations for average humidity environments are in good
389 agreement with the CNR DT 200 (2004) reduction factors corresponding to internal/external
390 conditions. However, the predicted degradation in the high relative humidity environment
391 conditions are much higher than the reduction factors proposed in the code for aggressive
392 environments. It is also noted that over factors, such as creep, fatigue or salt crystallization, are
393 not considered but they can have an effect on further reducing the bond strength.

394

395 **Conclusions**

396 The results of an extensive experimental program aimed at investigating the durability of FRP-
397 masonry systems were presented in this study. Accelerated ageing tests were performed
398 following two different hygrothermal conditions consisting of thermal cycles from +10°C to
399 +50°C with 90% R.H., called HT1, and 60% R.H., called HT2. The HT1 exposure was used for
400 simulating the thermal variations in wet environments, while HT2 simulated environments with

401 average relative humidity. The bond degradation was studied by visual inspection and single-lap
402 shear bond tests. The changes in mechanical properties of material constituents were also
403 investigated. Based on the experimental data, a degradation model was finally used to predict the
404 long-term performance of the studied system. Based on the obtained results, the following
405 conclusions can be drawn:

406 • The hygrothermal exposures did not affect the mechanical properties of the bricks.
407 However, epoxy resin and GFRP coupons showed some degradation. Generally, higher
408 degradation levels were observed due to exposing the specimens to HT1 conditions.

409 • FRP delamination was observed at the FRP/brick interface after exposure to
410 environmental conditions. The delamination, being due to the thermal incompatibility between
411 brick and adhesive, was progressively increased with the number of cycles. Moreover,
412 significantly larger FRP delaminations with higher growth rates were observed in the specimens
413 exposed to HT1 conditions. This can be due to the effect of moisture on the debonding fracture
414 energy and adhesive fracture properties.

415 • A progressive degradation of bond strength and fracture energy was observed in the
416 specimens. The degradation in the specimens exposed to HT2 was very small, contrarily to the
417 large reductions observed in the specimens exposed to HT1 conditions. In HT1 exposure, the
418 failure mode of the specimens changed progressively from cohesive failure in the brick to
419 adhesive failure at the FRP-brick interface due exposure time. However, no significant change of
420 failure mode was observed in the specimens exposed to HT2 conditions.

421 • An exponential predictive model was finally used for modeling the observed degradation
422 in the material properties and bond behavior. The models, once validate, were used for long-term

423 performance assessment of FRP-strengthened masonry elements and the obtained degradation
424 levels were compared with the reduction factors proposed in the current design guidelines.

425

426 **Acknowledgements**

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428 (Fundação de Ciência e Tecnologia, FCT), through grant SFRH/BD/80697/2011.

429

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Table 1. Material properties (five specimens).

Clay brick		Average	CoV(%)
Compressive strength	f_{cb} (MPa)	14.3	4.0
Flexural tensile strength	f_{tb} (MPa)	1.5	24.6
Epoxy resin			
Tensile strength	f_{te} (MPa)	53.8	9.7
Elastic modulus	E_{te} (GPa)	2.5	9.5
Ultimate strain	ε (%)	2.6	10.4
Primer			
Tensile strength	f_{tp} (MPa)	55.3	11.1
Elastic modulus	E_{tp} (GPa)	2.9	6.1
Ultimate strain	ε (%)	2.1	14.5
GFRP coupon			
Tensile strength	f_{tf} (MPa)	1250	15.0
Elastic modulus	E_{tf} (GPa)	79.2	6.8
Ultimate strain	ε (%)	1.9	20.2

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Table 2. Error in degradation modeling in HT1 exposure.

Cycles	Epoxy				GFRP				Bond			
	f_{te}/f_{te0}	Err. (%)	E_{te}/E_{te0}	Err. (%)	f_{tf}/f_{tf0}	Err. (%)	E_{tf}/E_{tf0}	Err. (%)	P/P_0	Err. (%)	G_f/G_{f0}	Err. (%)
0	0.99	-1.1	1.00	-0.1	1.00	-0.2	0.99	-0.5	1.08	7.6	1.00	-0.3
60	0.95	5.4	0.97	0.2	0.89	4.1	0.82	-1.1	0.87	-10.1	0.82	-17.7
120	0.91	-4.4	0.96	0.9	0.83	-3.9	0.76	3.0	0.73	-11.9	0.69	-25.6
180	0.89	-6.2	0.95	-1.8	0.80	3.3	0.75	1.7	0.63	18.2	0.58	46.4
225	0.87	8.8	0.94	1.0	0.79	0.8	0.74	-3.9	0.58	5.4	0.51	30.0
Average Err.		5.2		0.8		2.5		2.1		10.6		24.0

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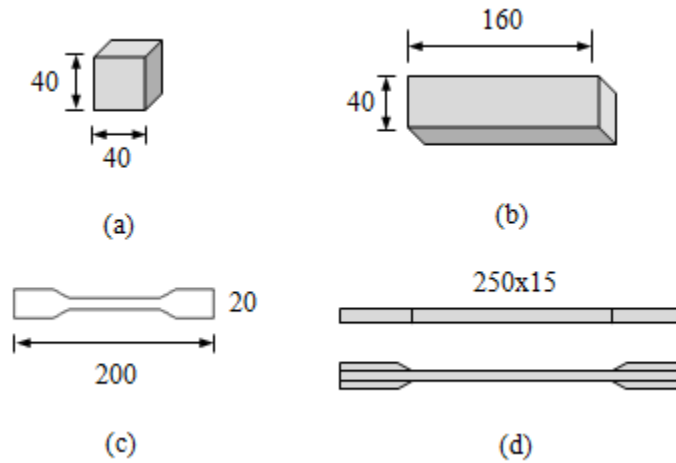
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Table 3. Error in degradation modeling in HT2 exposure.

Cycles	Epoxy				GFRP				Bond			
	f_{te}/f_{te0}	Err. (%)	E_{te}/E_{te0}	Err. (%)	f_{tf}/f_{tf0}	Err. (%)	E_{tf}/E_{tf0}	Err. (%)	P/P_0	Err. (%)	G_f/G_{f0}	Err. (%)
0	1.00	-0.1	1.00	-0.1	1.01	0.9	1.00	-0.1	1.02	1.8	1.05	4.8
120	0.93	0.4	0.94	2.6	0.93	-4.8	0.92	-4.0	0.89	-10.1	0.89	-13.4
180	0.91	-5.4	0.92	-7.6	0.90	1.8	0.89	10.5	0.86	2.3	0.85	-2.8
250	0.91	3.7	0.90	4.6	0.88	1.7	0.87	-4.1	0.83	4.9	0.81	17.1
360	0.90	3.2	0.88	6.7	0.87	-2.2	0.84	-4.5	0.81	-12.8	0.79	-19.8
480	0.90	-2.5	0.87	5.7	0.86	2.7	0.82	3.3	0.80	6.2	0.77	8.5
600	0.90	-0.8	0.86	5.8	0.85	4.5	0.81	2.1	0.79	-5.9	0.77	-14.7
710	0.90	-1.9	0.86	0.0	0.85	-2.9	0.80	-6.0	0.79	-3.7	0.76	-3.7
820	0.90	-0.9	0.85	0.7	0.85	-4.7	0.80	1.7	0.79	-9.2	0.76	-21.2
Average Err.		2.1		3.7		2.9		4.0		6.4		11.8

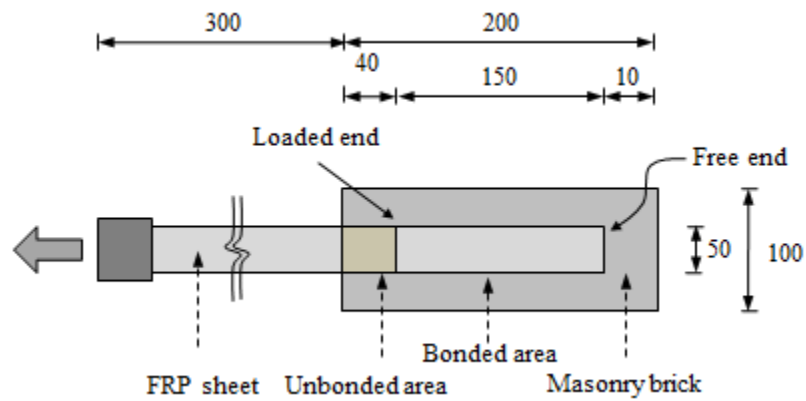
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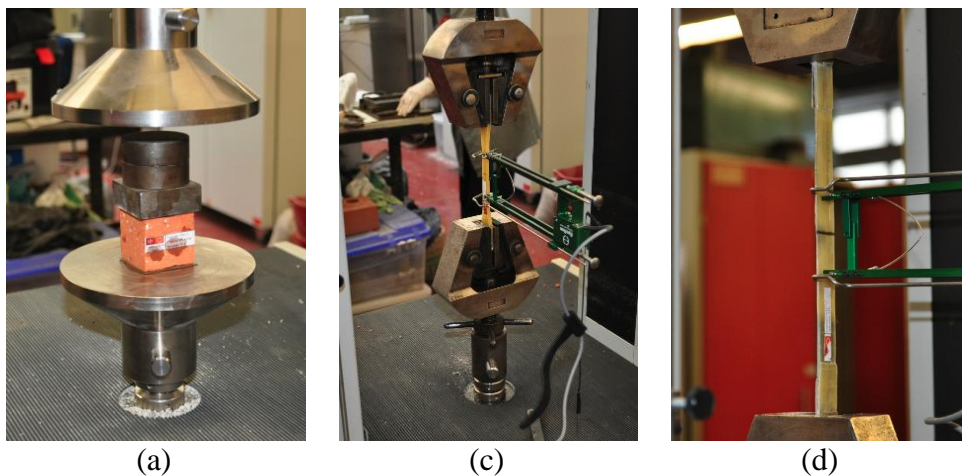
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Fig. 1. Specimens used for material testing (dimensions in mm): (a) brick cubic specimen; (b) brick prism; (c) epoxy resin and primer; (d) GFRP coupon.



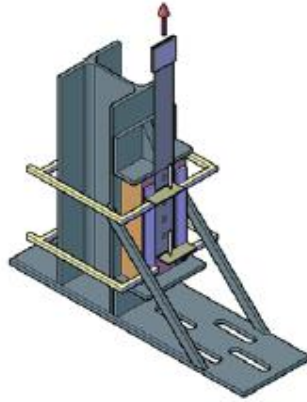
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Fig. 2. Geometry of specimens prepared for bond tests (dimensions in mm).



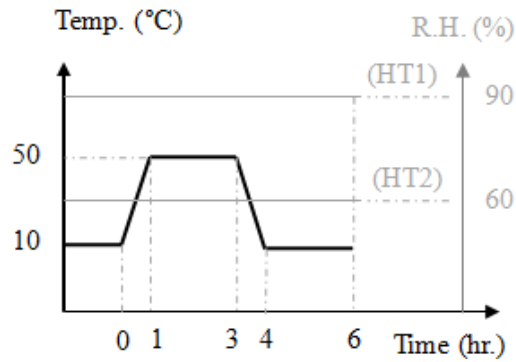
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Fig. 3. Mechanical characterization test setups: (a) brick compressive test; (b) epoxy tensile test; (c) GFRP tensile test.



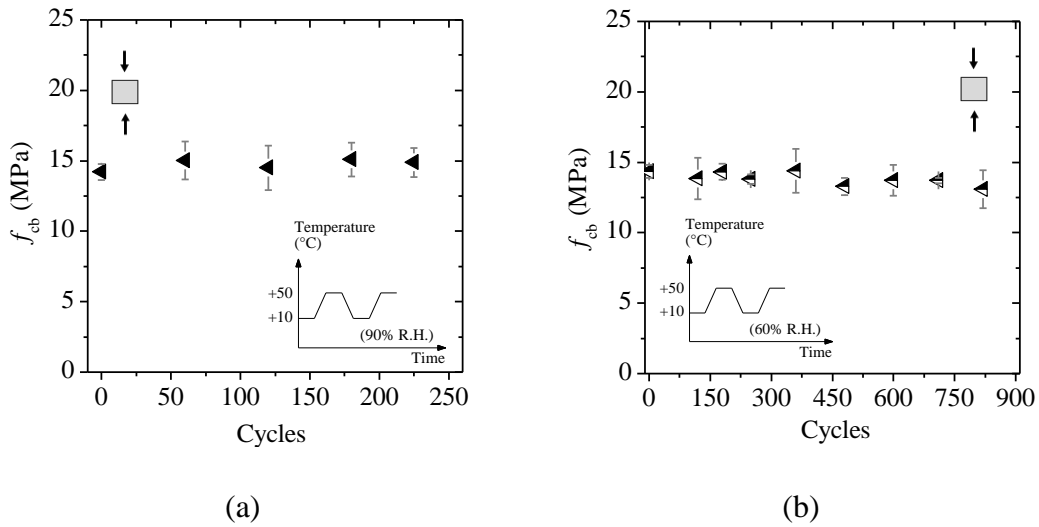
586 Fig. 4. Single-lap shear bond test setup (perspective).

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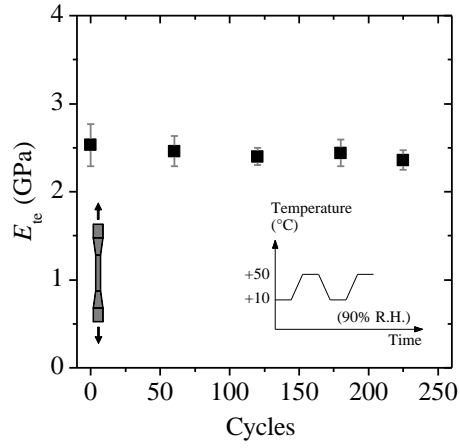
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Fig. 5. Hygrothermal exposures.

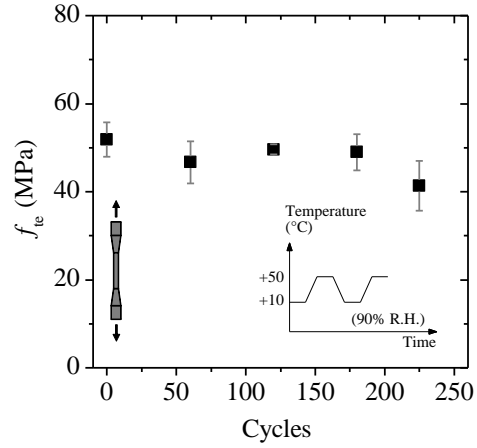


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Fig. 6. Degradation of bricks compressive strength due to exposures: (a) HT1; (b) HT2.



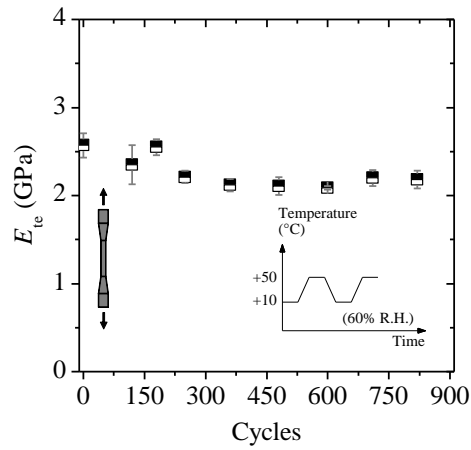
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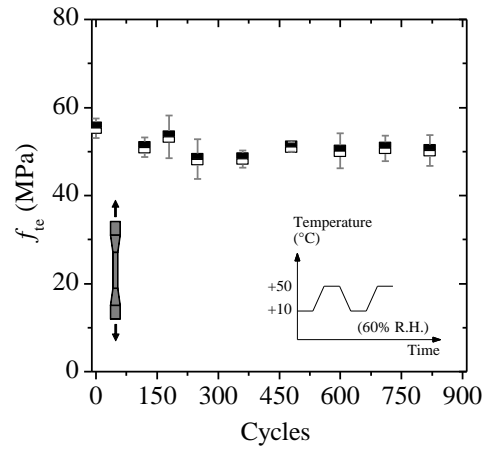
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Fig. 7. Effect of exposure HT1 on epoxy resin: (a) elastic modulus; (b) tensile strength.



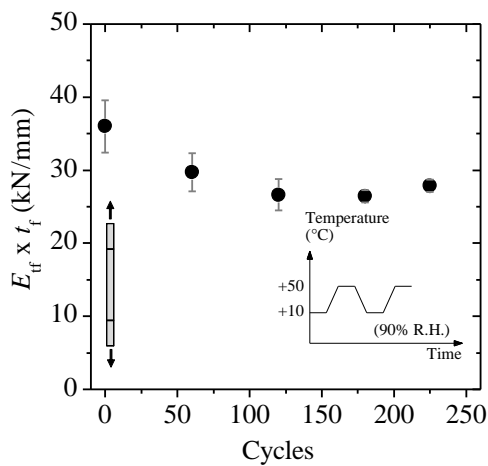
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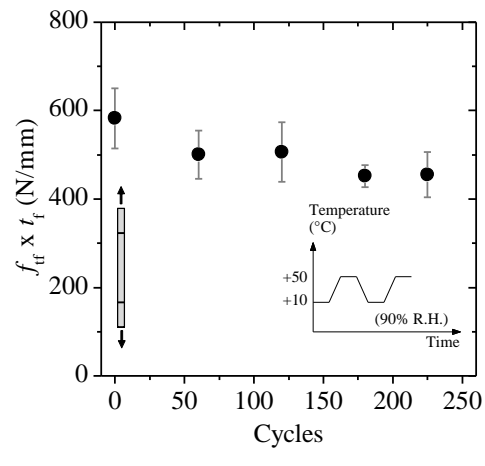
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Fig. 8. Effect of exposure HT2 on epoxy resin: (a) elastic modulus; (b) tensile strength.



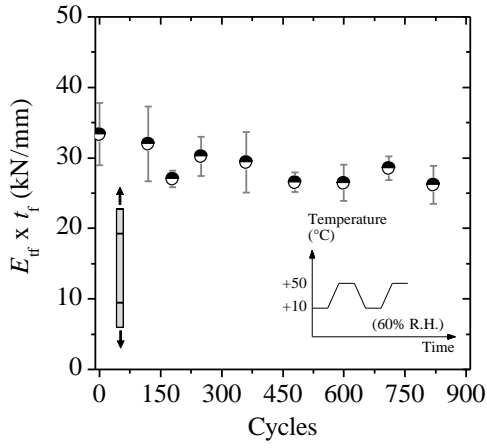
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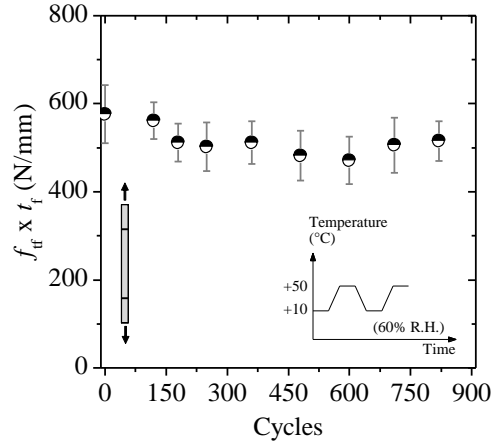
(b)

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Fig. 9. Effect of HT1 exposure on GFRP coupons: (a) elastic modulus; (b) tensile strength.

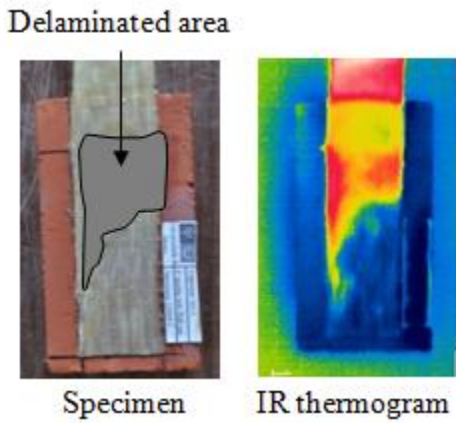


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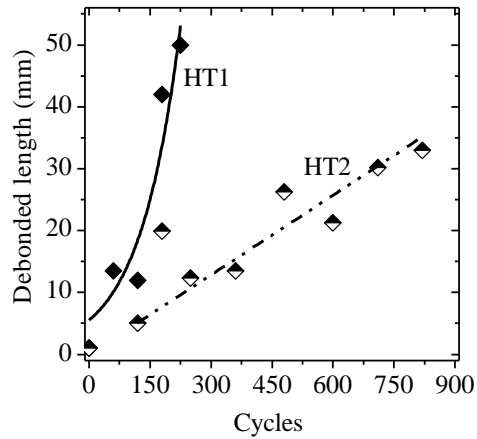


(b)

598 Fig. 10. Effect of HT2 exposure on GFRP coupons: (a) elastic modulus; (b) tensile strength.
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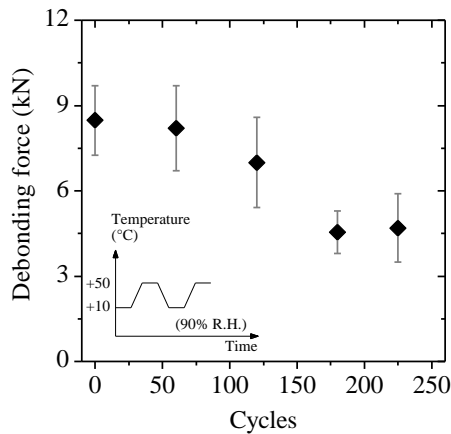


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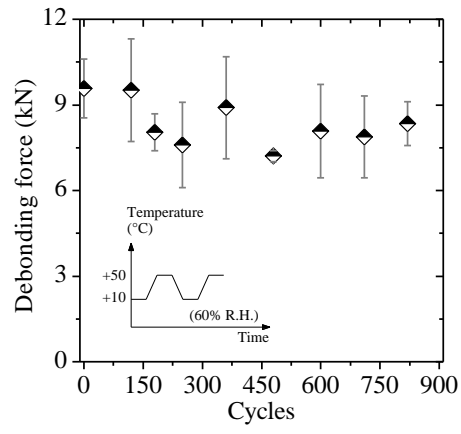


(b)

600 Fig. 11. Hygrothermal induced delaminations: (a) IR thermography results; (b) debonded length
601 growth.
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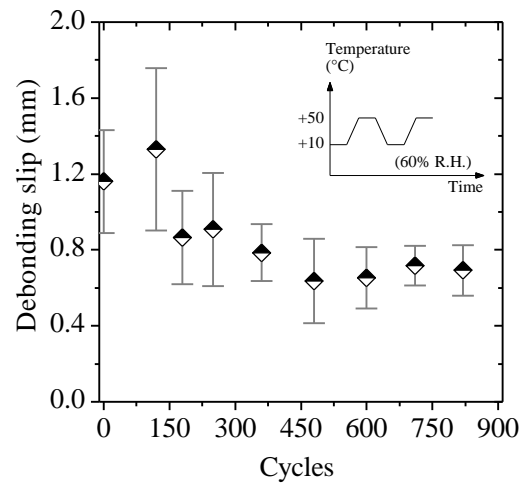
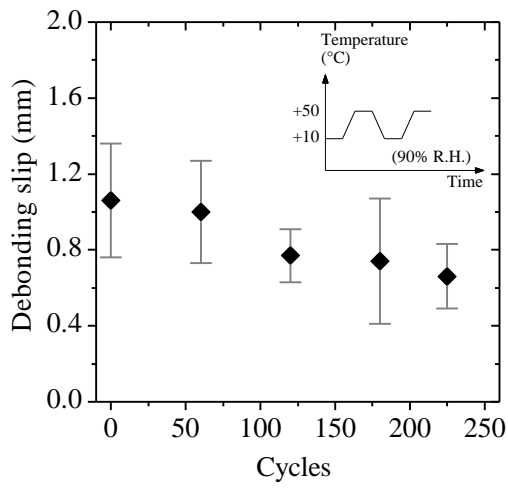


(a)



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603 Fig. 12. Debonding force degradation due to exposure: (a) HT1; (b) HT2.



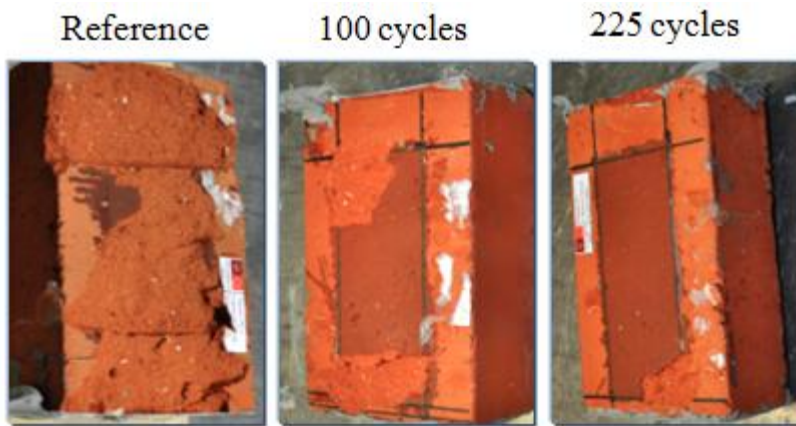
(a)

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Fig. 13. Changes in debonding slip due to exposure: (a) HT1; (b) HT2.

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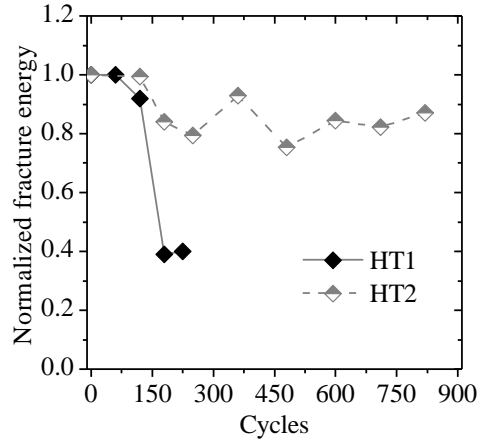


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Fig. 14. Change of failure mode in the specimens during HT1 exposure.

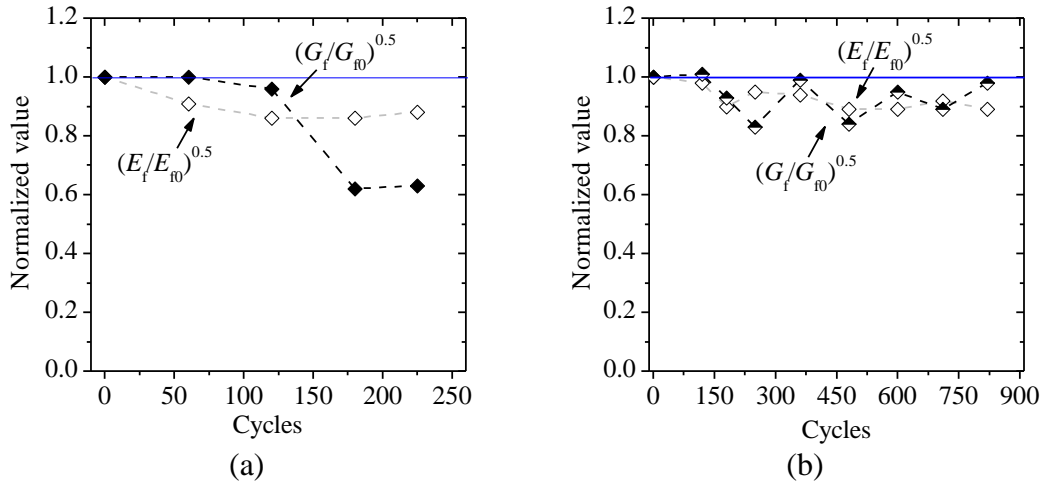


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Fig. 15. Degradation in the fracture energy.

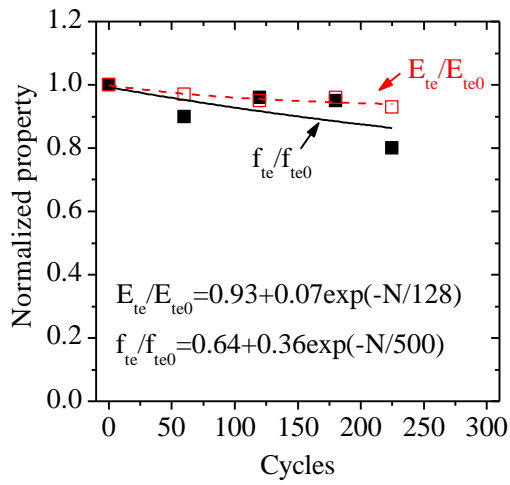
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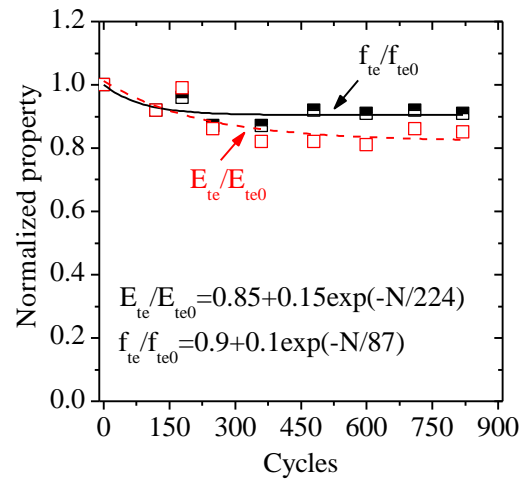
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Fig. 16. Bond degradation mechanisms in: (a) exposure HT1; (b) exposure HT2.

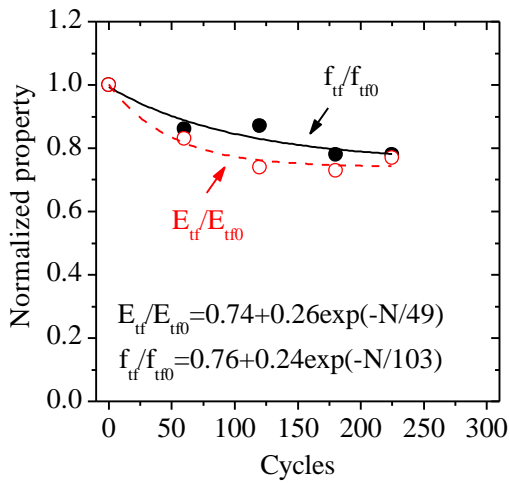
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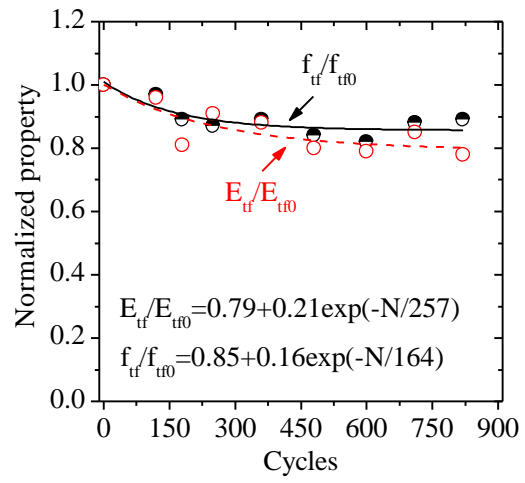
(a)



(b)



(c)



(d)

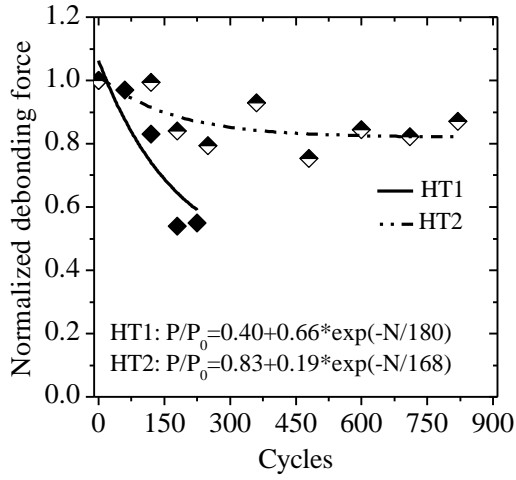
614 Fig. 17. Modeling the degradation in: (a) epoxy resin in HT1 environment; (b) epoxy resin in
 615 HT2 environment; (c) GFRP in HT1 environment; (d) GFRP in HT2 environment.

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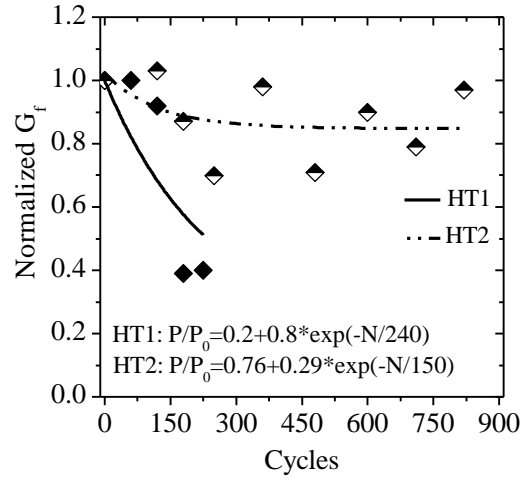
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(a)



(b)

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Fig. 18. Modeling the degradation of bond: (a) debonding force; (b) fracture energy.

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